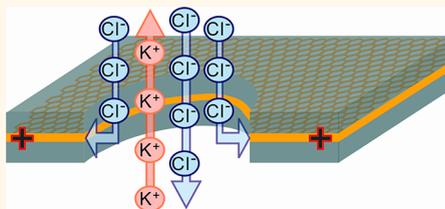


# Electrochemistry at the Edge of a Single Graphene Layer in a Nanopore

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**ABSTRACT** We study the electrochemistry of single layer graphene edges using a nanopore-based structure consisting of stacked graphene and Al<sub>2</sub>O<sub>3</sub> dielectric layers. Nanopores, with diameters ranging from 5 to 20 nm, are formed by an electron beam sculpting process on the stacked layers. This leads to a unique edge structure which, along with the atomically thin nature of the embedded graphene electrode, demonstrates electrochemical current densities as high as  $1.2 \times 10^4$  A/cm<sup>2</sup>. The graphene edge embedded structure offers a unique capability to study the electrochemical exchange at an individual graphene edge, isolated from the basal plane electrochemical activity. We also report ionic current modulation in the nanopore by biasing the embedded graphene terminal with respect to the electrodes in the fluid. The high electrochemical specific current density for a graphene nanopore-based device can have many applications in sensitive chemical and biological sensing, and energy storage devices.



**KEYWORDS:** Nanopores · graphene · graphene electrochemistry · nanobio sensors · stacked graphene

Graphene has attracted tremendous interest in the scientific world over the recent years due to its unique electronic,<sup>1,2</sup> thermal,<sup>3</sup> and optical<sup>4</sup> properties. It has shown great promise in the field of electronics, biological and chemical sensing, and energy storage applications.<sup>5,6</sup> Studies on graphene electrochemistry have suggested the ability of graphene-based electrodes to carry a large amount of current at electron transfer rates superior to graphite and carbon nanotube (CNT) electrodes.<sup>5</sup> The relative abundance of carbon on earth combined with widespread knowledge of carbon-based chemistries and stability makes the study of graphene-based electrochemistry extremely exciting.<sup>5,7,8</sup>

Graphene is a single atom thick sheet of sp<sup>2</sup> hybridized carbon atoms arranged in a honeycomb lattice structure. A graphene sheet has two types of electron transfer sites—edge and basal. Edge sites have already been demonstrated to possess enhanced electron transport rates and reactivity in studies of CNT ends.<sup>9</sup> Graphene has a higher theoretical specific surface area (2630 m<sup>2</sup>/g) than graphite and CNTs (1315 m<sup>2</sup>/g) and provides motivation for

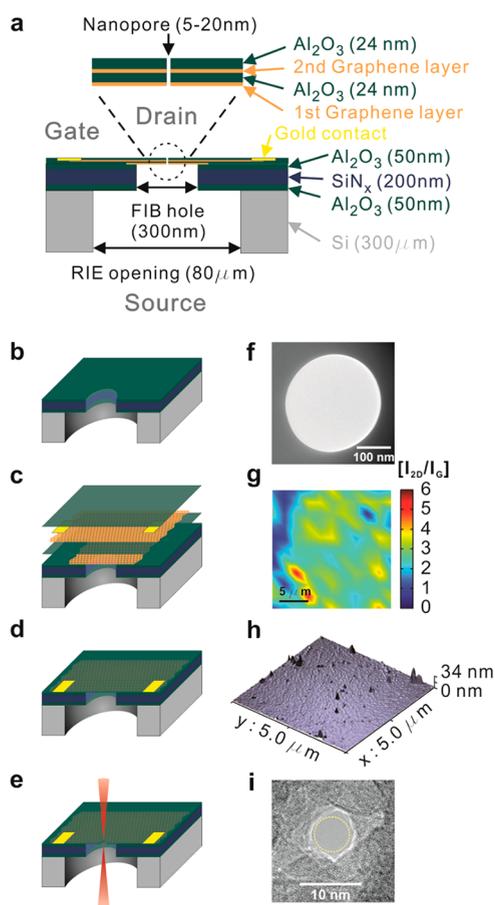
study of heterogeneous electron transfer rates.<sup>8</sup> In addition, graphene can carry significant current densities without degradation from electro-migration which typically causes significant damage in ultrathin metal films.<sup>10</sup> Current densities as high as  $2 \times 10^9$  A/cm<sup>2</sup> have been reported for nanoscale interconnects based on graphene grown by chemical vapor deposition (CVD).<sup>11</sup> The graphene edge plane atoms have been reported to have significantly higher electron transfer rates compared to basal planes in electrochemical studies on both highly ordered pyrolytic graphite as well as on multiple layers of graphene.<sup>12,13</sup> Graphene-modified glassy carbon electrodes have been reported to have much greater electrochemical response than unadulterated glassy carbon electrodes to molecules like paracetamol, hydrazine, glucose, and ethanol dopamine as well as heavy metals.<sup>7,8</sup> Zhou *et al.*<sup>14</sup> demonstrated the ability of chemically reduced graphene oxide electrodes to distinguish the electrochemical current signal from the four bases of DNA, which could not be distinguished with graphite and glassy carbon electrodes. Another important application of graphene electrochemistry is in energy storage devices. The specific

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**Figure 1.** Schematic diagram of graphene-embedded stacked membrane structure and fabrication. (a) Schematic showing the thickness of each layer as well as diameters of RIE, FIB, and nanopore holes; (b) supporting membrane consists of three layers of 50 nm of  $\text{Al}_2\text{O}_3$ , 200 nm of  $\text{SiN}_x$ , and 50 nm of  $\text{Al}_2\text{O}_3$ , deposited on 300  $\mu\text{m}$ -thick double polished prime Si wafer. RIE is used to etch an 80  $\mu\text{m}$ -wide opening in the Si wafer up to the supporting membrane and a 300 nm-through hole is fabricated in the supporting membrane by FIB. (c) The first graphene layer transferred onto the FIB hole acts as the support for subsequent layers. This is insulated from the second graphene layer by 24 nm of  $\text{Al}_2\text{O}_3$  deposition. The second graphene layer, which is the active electrode at the middle of membrane, is transferred onto the first  $\text{Al}_2\text{O}_3$  layer. Ti/Au deposition enables the formation of contacts. A further layer of  $\text{Al}_2\text{O}_3$  is deposited to insulate the electrode from the ionic solution. (d) Final structure of graphene embedded membrane suspended on 300 nm FIB hole. (e) Focused electron beam (CBED mode) in TEM is used to fabricate a single nanopore of 5–20 nm diameter. (f) TEM image of FIB hole of 300 nm diameter in supporting membrane. (g) Raman spectroscopy map of the  $I_{2D}/I_G$  ratio obtained from graphene surface indicating predominantly monolayer coverage. (h) AFM image of membrane surface. Roughness ( $R_a = 1.89 \pm 0.67$  nm) is significantly reduced on deposition of  $\text{Al}_2\text{O}_3$  on graphene compared to a bare graphene surface ( $R_a = 0.84 \pm 0.21$  nm). (i) A 5 nm nanopore is fabricated by convergent electron beam in TEM.

capacitance of chemically modified graphene was found to be up to 1352 F/g,<sup>5</sup> and extremely high energy densities up to 85.6 Wh/kg at room temperature have been reported.<sup>15,16</sup> Furthermore, graphene and hybrid graphene-based electrodes have been used to increase specific capacities of  $\text{Li}^+$  ion based

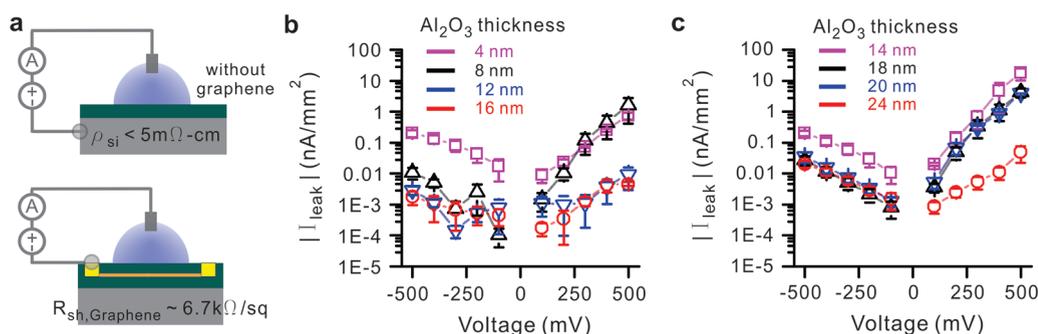
batteries, improving power density and cyclic performance, while maintaining mechanical integrity at high current densities.<sup>6</sup>

Despite extensive studies on graphene sheets and graphene doped electrodes, the electrochemical properties of isolated graphene edges remain relatively unexplored. Here, we demonstrate a graphene edge embedded nanopore (GEEN) structure to isolate graphene edge electrochemical activity from basal plane activity. Transmission electron microscopy (TEM) based sculpting offers potential for control on graphene edge structures.<sup>17</sup> Furthermore, we demonstrate the use of the embedded graphene edge to modulate the ionic flux in the nanopore. Along with a conductive graphene terminal of thickness equivalent to the distance between two adjacent base pairs in dsDNA ( $\sim 0.34$  nm), this could provide a basis for single DNA molecule analysis with measurement methodologies like tunneling or electrochemical redox reactions.

## RESULTS AND DISCUSSION

The fabrication of graphene nanopores using a TEM has been demonstrated previously and used to sense biomolecules like polynucleotides and DNA protein complexes.<sup>18,19</sup> In this study, we fabricate GEENs in stacked graphene and dielectric layers using a focused electron beam in a TEM (200 keV), and measure the electrochemical current exchange at the graphene edge embedded within the nanopore. The top  $\text{Al}_2\text{O}_3$  dielectric layer isolates the basal plane electrochemical activity. We demonstrate the very high electrochemical current density as well as the first known study of electrochemical current exchange at the graphene (potentially as thin as single layer) edge in an ionic solution. The combination of nonlinear diffusion at nanoscale electrodes, an enhanced concentration gradient of ions in the vicinity of the nanopore,<sup>20</sup> and high electron transfer rates at damaged edges of graphene<sup>12</sup> creates a unique system with high electrochemical current densities.

The schematic of our test GEEN structures is shown in Figure 1a.<sup>21</sup> The fabrication process is further described in Figure 1b–e (details in the Experimental Details section). Initially, a suspended hydrophilic supporting membrane of stacked layers of 50 nm  $\text{Al}_2\text{O}_3$ , 200 nm  $\text{SiN}_x$ , and 50 nm  $\text{Al}_2\text{O}_3$  is fabricated using deep reactive ion etching (DRIE). Subsequently, a hole of  $300 \pm 40$  nm is formed in the supporting membrane using a focused ion beam (FIB) (Figure 1b,f). The graphene– $\text{Al}_2\text{O}_3$  stack is then formed on the supporting membrane with the FIB hole by transferring graphene films grown by CVD (see Experimental Details section). We note that the hydrophilic nature of the supporting membrane helps spread the water more evenly during the graphene transfer steps and improves the smoothness of the transferred graphene/PMMA stack.<sup>22</sup> The Raman spectroscopy maps of the graphene 2D to G



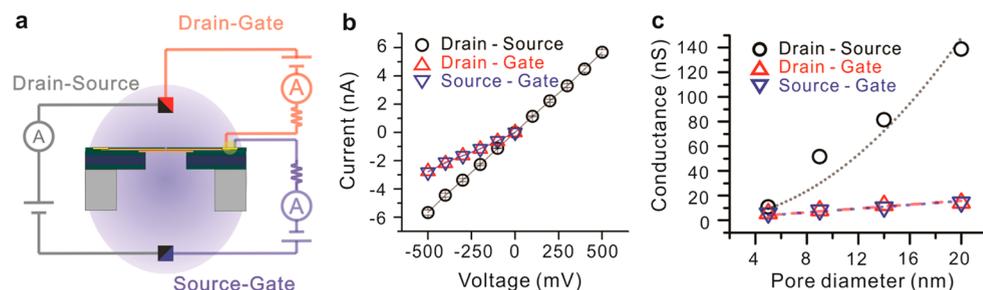
**Figure 2.** Leakage test on various thicknesses of  $\text{Al}_2\text{O}_3$ . (a-top) Schematics showing leakage measurement setup for  $\text{Al}_2\text{O}_3$  on p++ silicon ( $\rho < 5 \text{ m}\Omega\text{-cm}$ ).  $\text{Al}_2\text{O}_3$  of thickness 4 to 16 nm was deposited on the conductive Si wafer. Measurements are conducted with one electrode connected to Si wafer and the other attached to Ag/AgCl electrode in the solution (a-bottom). Schematic of leakage measurement setup for  $\text{Al}_2\text{O}_3$  on graphene transferred onto Si surface with  $\text{Al}_2\text{O}_3$  deposited on top.  $\text{Al}_2\text{O}_3$  thickness in the range of 14–24 nm is deposited on graphene ( $R_{\text{sh}} \approx 6.7 \text{ k}\Omega/\text{sq}$ ) transferred on a Si wafer with an ALD-deposited  $\text{Al}_2\text{O}_3$  top surface. Measurements are conducted between the graphene film contacted with aluminum wires and the solution contacted with Ag/AgCl electrodes. All leakage experiments are performed in 1 M KCl, 10 mM Tris, 1 mM EDTA at pH 7.6, and at room temperature ( $22 \pm 2 \text{ }^\circ\text{C}$ ). (b) Leakage current density measured for  $\text{Al}_2\text{O}_3$  on conductive silicon.  $\text{Al}_2\text{O}_3$  thickness less than 10 nm showed leakage current greater than  $1 \text{ nA}/\text{mm}^2$  at 500 mV, but thicker  $\text{Al}_2\text{O}_3$  ( $>10 \text{ nm}$ ) showed much greater insulation over the voltage range of  $-500$  to  $+500 \text{ mV}$ . (c) Leakage current density for  $\text{Al}_2\text{O}_3$  deposited on graphene. Leakage current is observed to be fairly high up to 20 nm-thick  $\text{Al}_2\text{O}_3$ . Also the leakage is significantly higher for positive voltage at Ag/AgCl electrode.  $\text{Al}_2\text{O}_3$  at 24 nm thickness displays decent insulation from leakage. Current leakage occurrence at relative thicker  $\text{Al}_2\text{O}_3$  deposited on graphene is associated with wrinkles on graphene (Supporting Information, Figure S9).

peak intensity ratios ( $I_{2D}/I_G$ ) (Figure 1g and Supporting Information, Figure S4a,b) and the full-width at half-maximum of the 2D peak ( $\text{fwhm}_{2D}$ )<sup>23</sup> (Supporting Information, Figure S4e–f) show our growth process results in a mix of monolayer and bilayer graphene, similar to our previous work.<sup>21</sup> The first graphene layer (G1) in our stack spans the FIB hole and acts as a mechanical support for deposition of the subsequent graphene and dielectric layers of our architecture. We note that subsequent to the graphene transfers, the membranes are annealed in an Ar/ $\text{H}_2$  atmosphere at  $400 \text{ }^\circ\text{C}$  to remove PMMA residue remnant from the transfer process.<sup>24</sup>

To ensure uniform nucleation of the subsequent  $\text{Al}_2\text{O}_3$  deposition (D1) onto the chemically inert graphene basal planes, a metallic seed layer of Al (2 nm thick) is evaporated onto the graphene.<sup>25</sup>  $\text{Al}_2\text{O}_3$  is a suitable choice as the dielectric because of its excellent mechanical stability<sup>26</sup> and reduction in  $1/f$  noise compared to  $\text{Si}_3\text{N}_4$  and  $\text{SiO}_2$  membranes.<sup>27,28</sup> The atomic force microscopy (AFM) images (Figure 1h and Supporting Information, Figure S4d) clearly show dense and uniform deposition of the dielectric due to the presence of the seed layer (Supporting Information, Figures S5a–c and S7) as compared to dielectric deposition without the Al seed layer (Supporting Information, Figure S5a,b). ALD is chosen as it offers subnanometer control over dielectric thickness in addition to being a conformal deposition technique and a low temperature process, making it compatible with the previously transferred graphene layers.<sup>21</sup> The thickness of the dielectric deposited is 24 nm, a value established through extensive leakage testing in fluidic environments (Figure 2 and Supporting Information, Figure S1). Similar thicknesses of dielectric have

been reported to provide effective isolation in ionic fluid environments in transistor-based devices.<sup>29,30</sup> A second graphene layer (G2) is transferred onto D1 and annealed in an Ar/ $\text{H}_2$  atmosphere. This layer is contacted using Ti/Au contacts and insulated by depositing another 24 nm of  $\text{Al}_2\text{O}_3$  (D2) as described above (see Experimental Details section).

To explore the electrochemical current exchange at the graphene nanopore edges, it is essential to eliminate current exchange at the basal plane from affecting our measurements. In the embedded graphene membrane, the parasitic leakage current from gate to source and gate to drain (indicated in Figure 1a) could adversely affect our experimental values. The active device area exposed to fluid on the backside (gate-source path) is just the area exposed to the FIB hole of 300 nm. This area is insulated from the fluid by the 24 nm  $\text{Al}_2\text{O}_3$  under the graphene gate electrode. The rest of the graphene is well insulated by a total thickness of 300 nm of stacked  $\text{Al}_2\text{O}_3$  and  $\text{SiN}_x$  layers of the supporting membrane structure. On the gate-drain path the entire encapsulated graphene sheet is shielded from the fluid by just the top layer of 24 nm  $\text{Al}_2\text{O}_3$ . The fluid area exposed at the top layer corresponds to the area exposed by the o-rings (diameter = 1.42 mm) used to seal the fluidic setup. To mimic our device structure and characterize leakage through the top dielectric, we fabricated the device as shown in Figure 2a. We compared the leakage current through different thicknesses of  $\text{Al}_2\text{O}_3$  deposited on a bare conductive silicon wafer and  $\text{Al}_2\text{O}_3$  deposited on a graphene sheet transferred onto an  $\text{Al}_2\text{O}_3$  coated (on Si wafer) top surface, similar to the D1/G2/D2 stack of our actual devices. The ALD dielectric deposition of  $\text{Al}_2\text{O}_3$  on graphene is preceded by the seed layer



**Figure 3.** Electrochemical measurements for embedded graphene nanoelectrode. (a) Schematic diagram of measurement setup. For the drain–source measurement (gray), the source is connected to ground and voltage applied at the drain. For drain–gate (red) and drain–source (blue) measurements, the gate is connected to ground and voltage is applied to the other terminal. (b) Current–voltage curve of nanopore ionic current and electrochemical behavior of graphene edge through 5 nm nanopore. Identical currents through the drain–gate and source–gate pathways indicate electrochemical exchange at the exposed graphene edge. (c) Conductance dependence on pore diameter. Drain–source conductance shows a square dependence on pore diameter, while gate current exchange shows a fairly linear dependence on pore diameter consistent with electrochemical exchange at cylindrical nanopore wall. The slight variation from linear dependence may be attributed to varying graphene sheet thickness on various regions of the membrane; 5, 9, 14, and 20 nm diameter nanopores were used in this study. All experiments are performed in 1 M KCl, 10 mM Tris, 1 mM EDTA at pH 7.6.

Al ( $\sim 2$  nm thick) evaporation as described previously. The leakage is measured by attaching a PDMS well (2.75 mm in diameter) on top of the device to hold the fluid. The current is measured between the graphene electrode and Ag/AgCl electrode dipped in the electrolyte fluid. All leakage measurements were performed with a 1 M KCl solution. The conductive silicon and the graphene electrode are connected to ground in all measurements.

The leakage densities observed are presented on a logarithmic scale (absolute value) in Figure 2b,c. On the bare silicon wafer, a slight asymmetry was observed in the  $I$ – $V$  characteristics. For a positive Ag/AgCl electrode voltage a higher leakage current density was observed through the dielectric. The leakage current density reduces from  $-0.2$  to  $-0.001$  nA/mm<sup>2</sup> at  $-500$  mV, as the dielectric thickness is increased from 4 to 16 nm. Comparing these values to leakage currents on samples with the dielectric deposited on graphene, we can see a significant increase in the leakage current of the latter (Figure 2c). The electrochemical exchange at the dielectric–electrolyte interface has been reported in electrolyte–oxide–silicon (EOS) devices.<sup>31</sup> Since the leakage current is high at positive electrode voltages, this could indicate electron tunneling through the pinholes in the dielectric as shown in our AFM images (Supporting Information, Figure S4d and Figure S9), similar to those reported in TiO<sub>2</sub>-coated CVD graphene membranes.<sup>18</sup> On the other hand, at negative electrode voltages the leakage currents are significantly suppressed in the voltage range from 0 to  $-500$  mV. Increasing the dielectric thickness from 14 to 24 nm decreases the leakage current density from  $-0.2$  to  $-0.02$  nA/mm<sup>2</sup>. For a 2.75 mm diameter PDMS well, that translates to a current of about 118.7 pA. Since the ionic current through the nanopore is usually in the range of nanoamperes, at least 1 order of magnitude lower leakage current is essential to maintain reliability of our electrochemical current measurements and to

have a gate current independent from interference due to leakages. Therefore, we use only the negative voltage range (0 to  $-500$  mV) in our nanopore measurements to minimize and avoid leakage across D2.

After settling upon a dielectric thickness of 24 nm, nanopores are drilled in this stacked structure using convergent beam electron diffraction (CBED) mode in a TEM (Figure 1e,i). We fabricated four different pore diameters (5, 9, 14, 20 nm) for our test structures. For a 5 nm pore, the beam sputters through the membrane in about 30s. For larger pores, sculpting is needed by moving the beam on the edges of the pore to expand it. Control is achieved by *in situ* monitoring of the nanopore dimension with imaging. Since TEM provides angstrom level precision we believe the nanopore dimensions are accurate within a tolerance of 1 nm. Prior to assembly in the fluidic setup the backside (silicon trench side) (Figure 1a) is O<sub>2</sub> plasma treated to make the pore hydrophilic to facilitate wetting.<sup>18,21</sup> The top graphene layer (G2) is contacted and the chip is encapsulated in a custom built fluidic setup (Supporting Information, Figure S6). Ethanol is then flushed into both chambers to promote wetting as reported in previous nanopore studies.<sup>19,21</sup> The ethanol is flushed away repeatedly with deionized (DI) water and the desired buffer solution is inserted into both chambers.

The schematics of drain–source, drain–gate, and source–gate measurements are shown in Figure 3a. An external resistor of 20 M $\Omega$  is placed in series with graphene. This helps ensure our graphene current measurements are not significantly affected by leakage. At 500 mV a 20 M $\Omega$  resistor conducts 25 nA of current. Since the currents observed are much less it indicates the electrochemical resistance at the graphene edge terminal is much higher and determines current in the series circuit. For a 1 M KCl solution used in these measurements, the drain–source conductance exhibits a squared dependence<sup>32</sup> with pore diameter

as indicated in Figure 3c. The current values for the different pore diameters also seem to be in good agreement with our previous work on similar structures.<sup>21</sup> For the source–gate and drain–gate measurements, the graphene gate is always connected to ground to maintain a positive voltage with respect to source or drain and ensure minimal leakage in accordance to our leakage measurements as described earlier. This is indicated in the  $I$ – $V$  curves for a 5 nm pore showed in Figure 3b (and Supporting Information, Figure S2). The drain–gate and source–gate conductance is also plotted in Figure 3c and is observed to be nearly identical for each of the four different pore diameters, indicating that the measured current is indeed only through the electrochemical exchange at the graphene terminal and the leakage contribution to these measurements on the drain side is indeed negligible. The o-rings used in these experiments are approximately  $1.42 \pm 0.1$  mm in diameter. On the basis of the leakage measurements, for a 24 nm thick  $\text{Al}_2\text{O}_3$  insulation layer, the maximum contribution of leakage at a drain/source at potential of  $-500$  mV should be approximately 30 pA, which is about 2 orders of magnitude less than the currents observed in these measurements. This is further confirmed by similar measurements in the same structure without a nanopore as currents in the range of 10–20 pA are observed across all three terminals. Furthermore, the conductance through the graphene terminal scales fairly linearly with pore diameter as seen in Figure 3c. The slight variation from the linear dependence can be explained from the varying graphene sheet thickness (Figure 1g) over the membrane, which affects the pore sidewall area, since the pore region could consist of a mixture of mono- and bilayer graphene. Nonetheless, we do see an increase of conductance from 5 to 15 nS as the pore diameter is increased from 5 to 20 nm. This is expected and indeed proves that this current is due to electrochemical exchange on the cylindrical pore sidewalls.

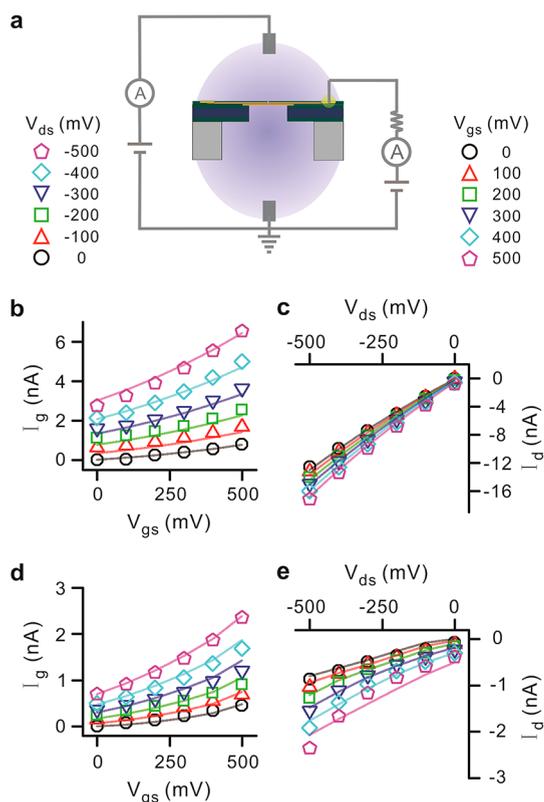
From the current values of electrochemical exchange at the 5 nm pore edge, (Figure 3b) and assuming a predominantly monolayer coverage of graphene, we calculate a current density of up to  $1.2 \times 10^4$  A/cm<sup>2</sup> at a drain voltage of  $-200$  mV. This current density is 3 orders of magnitude higher than electrochemical current densities reported for oxygen reduction on CNT electrodes.<sup>29</sup> From electrochemistry studies on basal planes of individual monolayer sheets for CVD-grown graphene reported by Li *et al.*,<sup>33</sup> a current density of about  $6 \times 10^{-8}$  A/cm<sup>2</sup> is obtained. Thus a significant electrochemical current enhancement is observed using individual graphene edges as the active electrode material. Furthermore, we simulated the concentration of  $\text{H}^+$  and  $\text{Cl}^-$  ions at the nanopore (see Experimental Details section) and the  $\text{Cl}^-$  ions are significantly higher in number. Thus, all redox couples

based on  $\text{H}^+/\text{OH}^-$  ions can safely be ignored as it is highly unlikely they can contribute to such high currents. Thus, we conclude that the reaction at the positive graphene electrode (anode) edge is the oxidation of  $\text{Cl}^-$  ions. The equilibrium oxidation potential for this reaction at room temperature is  $-1.36$  V.<sup>34</sup> However surface treatments enhancing the number of possible adsorption sites in diamond electrodes have been reported to lower the potential of chloride oxidation by as much as 0.5 V.<sup>35</sup> A similar mechanism might explain high electron transfer rates observed on graphene edges at low voltages. Electrochemical studies on graphite edges have exhibited extremely high electrochemical reaction rates.<sup>12,13</sup> Fast electron transfer kinetics reported on CNTs are also attributed to tube ends, identified as the reactive sites.<sup>9,36,37</sup> For GEENs we expect all sites at the nanopore edge to be damaged. Girit *et al.*<sup>17</sup> reported TEM drilled graphene nanopores which reconstruct and eventually exhibit a zigzag edge configuration due to its higher stability. For a graphene nanoribbon with zigzag edges, a large peak in the density of states is observed at the edges,<sup>38,39</sup> as confirmed by scanning tunnelling microscopy (STM) studies.<sup>40</sup> An enhancement in the density of states at the graphene nanopore edges of our architecture may have a direct effect in enabling the high electrochemical current densities observed in our measurements.

We note an electrochemical reaction consists of mass transport of the reactive species to the electrode surface and electron exchange at the electrode surface.<sup>41</sup> Since the dominant electrochemical exchange in our measurements occurs at the damaged graphene nanopore edges, it would appear that the electron exchange is not the rate limiting step. Diffusion limited electrochemical systems operate in the linear diffusion regime. For linear diffusion based systems, that is, when the electro-active length is comparable to the diffusion layer thickness, the reaction is diffusion limited and the peak current  $i_p$  is given by the Randles-Sevcik equation:<sup>9</sup>

$$i_p = 2.69 \times 10^5 n^{3/2} A C D^{1/2} v^{1/2} \quad (1)$$

where,  $n$  is the number of moles of electrons transferred in the reaction,  $A$  is the area of the electrode (cm<sup>2</sup>),  $C$  is the analyte concentration (in mol/cm<sup>3</sup>),  $D$  is the diffusion coefficient (cm<sup>2</sup>/s), and  $v$  is the scan rate (V/s) of the applied potential. For a chloride ion oxidation reaction,  $n$  is assumed to be 1. The active area of the electrode is the cylindrical pore area, which for a 5 nm pore, is calculated to be  $9.4 \times 10^{-14}$  cm<sup>2</sup>. The concentration is taken as  $10^{-3}$  mol/cm<sup>3</sup> and the diffusion coefficient of  $\text{Cl}^-$  is taken as  $1.5 \times 10^{-5}$  cm<sup>2</sup>/s.<sup>42</sup> For a 5 nm pore and a scan rate of 100 mV/10 s the peak current by the above equation gives  $i_p = 9.6 \times 10^{-7}$  nA, which is much smaller than the observed current.



**Figure 4.** Three terminal measurement for the graphene embedded membrane. (a) Schematic diagram with source connected to ground while voltage is swept at the drain and gate terminals. (b and d) Gate current characteristics for 1 M KCl and 10 mM KCl, respectively. The variation of gate current with gate source bias as drain voltage is varied is shown. The scatter points are experimental data while the straight lines are simulation fits. (c and e) Drain current characteristics for 1 M KCl and 10 mM KCl solution, respectively. The variation of drain current with drain source bias as gate voltage is varied is recorded. Both solutions are prepared with 10 mM Tris and 1 mM EDTA for buffering at pH 7.6.

Thus the reaction is not diffusion limited. It should be noted that our electrode size is in nanometers and is much smaller than the diffusion layer thickness (usually of the order of  $(Dt)^{1/2} = 3.8 \times 10^{-2} \text{ cm}$ ,<sup>43</sup> where  $t$  is the time period of each scan) and hence convergent diffusion effects are significant. For microelectrodes, convergent diffusion leads to significantly higher mass transport and thus higher current densities.<sup>9</sup> We believe that with the graphene nanoelectrodes used in our experiments, this effect would be exacerbated. Furthermore, the local concentration of the electro-active species ( $\text{Cl}^-$ ) is much higher and a 3-fold increase has been reported when compared to microelectrodes of same electro-active area. This increase in concentration in the vicinity of the nanopore as compared to the bulk solution results in faster mass transport,<sup>20</sup> contributing to the large current densities measured in our GEEN structures. Our simulations indicate a local (nanopore edge) concentration of  $\text{Cl}^-$  as high as 8.5 M for a bulk KCl concentration of 1 M (details in the Methods section).

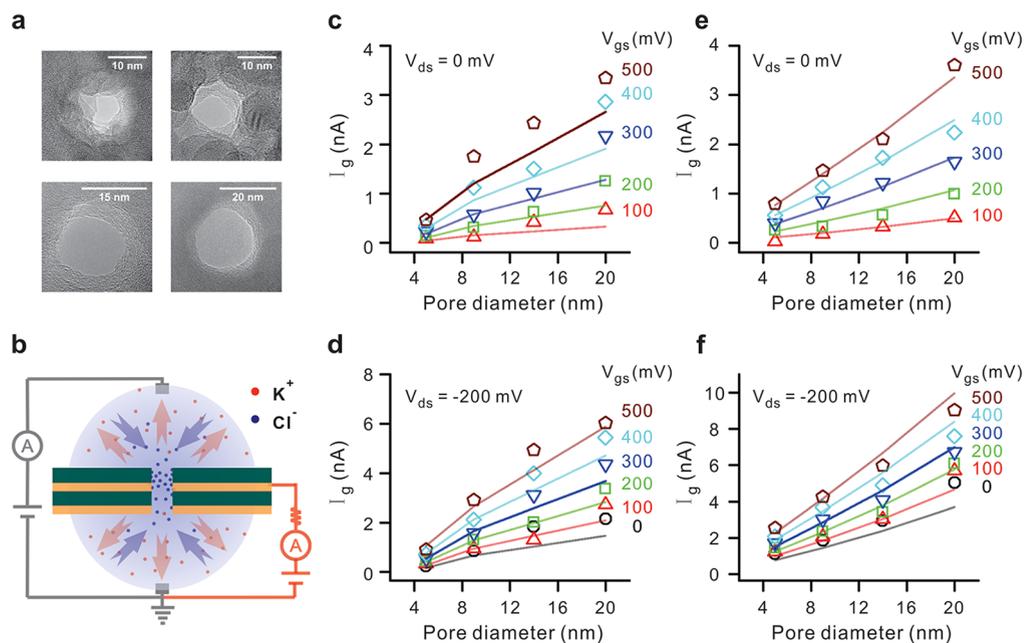
We further investigate the use of our structure as a three-terminal device analogous to a transistor (Figure 4a). The source terminal is always connected to ground in these measurements. The source current can be obtained by Kirchhoff's law

$$I_d = I_g + I_s \quad (2)$$

where,  $I_d$ ,  $I_g$ , and  $I_s$  are the drain, gate, and source currents, respectively. In accordance with our leakage results, the drain is always kept at a negative potential with respect to the gate for minimal interference from leakage. The graphene gate current characteristics ( $I_g$  vs  $V_{gs}$ ) for a 5 nm pore in 1 M KCl solution are shown in Figure 4b. A shift in the gate current values is observed as the drain voltage is swept from 0 to  $-500$  mV at a sweep rate of 100 mV/10 s (step function). Numerical simulations are used to explain gate current characteristics (see Experimental Details section). The  $I_g$  dependence on  $V_{gs}$  and  $V_{ds}$  voltage is estimated by an exponential function. This equation is coupled with the Poisson-Nernst-Planck equation and the Grahame equation and solved simultaneously to obtain both gate and drain current values. Figure 4c shows the  $I_d$  vs  $V_{gs}$  characteristics. As expected, a shift in the  $I_d$  is seen as  $V_{gs}$  is swept. The simulation results are in good agreement with the experimental data at 1 M KCl.

Further confirmation of observation of graphene edge currents is obtained by repeating these measurements for three more pores of 9, 15, and 20 nm diameters (Figure 5a). Since the measured currents are due to electrochemical exchange at the pore walls, the active area is cylindrical. Thus linear dependence of  $I_g$  on pore diameter is expected. We previously noted this in our two-terminal measurements for currents due to electrochemical exchange at graphene electrodes (Figure 3c). At 1 M KCl buffer solution the  $I_g$  values at two different  $V_{ds}$  (0 and  $-200$  mV) are shown in Figure 5e,f, respectively, for  $V_{gs}$  swept between 0 to  $+500$  mV. The simulated values (solid lines) show excellent agreement with experimental data (symbols) for all four pore diameters. We see a 4-fold increase in the  $I_g$  value as the pore size is increased from 5 to 20 nm.

Similar experiments were repeated in 10 mM KCl solution. The ( $I_g$  vs  $V_{gs}$ ) and ( $I_d$  vs  $V_{ds}$ ) characteristics show a similar shift as expected (Figure 4d,e). The simulation results (solid lines in both graphs) are in good agreement with the experimental results, although in this case the fitting parameters are altered for a 5 nm pore since the pore diameter is smaller than the Debye layer thickness (see Experimental Details section).<sup>44</sup> The  $I_g$  values do not scale linearly with concentration and this is attributed to enhanced ionic flux in the vicinity of the nanopore as shown in our simulations (see Experimental Details section). The pore diameter dependence measurements in 10 mM KCl for all four pore diameters show fairly good



**Figure 5.** Gate current dependence on pore diameter: (a) TEM images of nanopores of four different diameters (5, 9, 14, 20 nm) nanopores drilled through an embedded graphene membrane. (b) Schematic diagram of electrochemistry. The positive gate bias leads to attraction of chloride ions to the nanopore and repulsion of potassium ions. Red dots and arrows represent potassium ions while blue dots and arrows are for chloride ions. (c–f) Scaling of gate current with pore size at drain bias of 0 and  $-200$  mV for 4 different pore diameters. (c,d) Gate current dependence on pore diameter using 10 mM KCl solution. Linear dependence on pore diameter is observed over gate bias ranging from 0 to  $+500$  mV for both drain bias values. (e,f) Gate current dependence on pore diameter using 1 M KCl solution. Similar linear dependence on pore diameter is observed over the entire voltage range. The scatter points are experimental data while the straight lines are simulation fits. Both solutions are prepared with 10 mM Tris and 1 mM EDTA for buffering at pH 7.6.

agreement with simulated results and a linear increase of  $I_g$  with pore diameter is displayed (Figure 5c,d). Note that the values of  $I_d$  and  $I_g$  are observed to be nearly the same for these measurements at 10 mM, as illustrated in Supporting Information, Figure S3. From eq 2, this implies an extremely low  $I_s$ .

Our results confirm the observation of electrochemical exchange at the graphene edge, isolated from basal plane electrochemical activity. An array of GEENs could potentially be used to harness the extremely large value of energy density per unit mass. Methods like electron beam, nanoimprint lithography, or helium-based focused ion beam<sup>45</sup> techniques could be used to mass produce arrays of nanopores. Improvement in dielectric coverage of graphene by use of other materials like  $\text{HfO}_2$ <sup>46</sup> and different dielectric seed layer materials<sup>47</sup> like titanium or 3,4,9,10-perylene tetracarboxylic acid dianhydride (PTCDA) would significantly enhance the voltage range used in these experiments by reducing parasitic leakage through the dielectric, enabling higher current densities to be harnessed.

Furthermore, we note that the differential flux in ions on opposite sides of the nanopore could potentially have interesting applications in controlling the flux of biomolecules to be sensed through electrochemical exchange at the graphene edge. The differential ion flow rate could potentially be used to trap molecules within the pore allowing for electrical interrogation

using the conductive graphene terminal. Wanunu *et al.*<sup>48</sup> reported the use of salt gradients as a means to enhance DNA capture rates to increase throughput of the detection scheme. Another major biosensing application of an embedded conductive terminal in a solid state nanopore is with regards to DNA sequencing.<sup>49</sup> STM based studies have been demonstrated<sup>50</sup> to distinguish deoxynucleotide monophosphates (dNMPs) and partially sequence DNA oligomers by using tunneling current measurements. Tsutsui *et al.*<sup>51,52</sup> demonstrated tunnelling current measurements to distinguish bases in deoxynucleotide triphosphate (dNTP) molecules. If an embedded conductive terminal, *e.g.*, GEENs, can be combined with biological<sup>53</sup> or electronic<sup>54</sup> methods to slow DNA translocation rates, it could provide a pathway to DNA sequencing.

## CONCLUSION

In summary, we present the investigation of electrochemical current exchange at CVD-grown graphene edges within a nanopore. We demonstrate the ability of our graphene embedded nanopore structures to study electrochemistry at graphene edges isolated from electrochemical contributions of the basal plane. We observed electrochemical current densities on the order of  $10^4$  A/cm<sup>2</sup>, 3 orders of magnitude higher than those reported for carbon nanotubes and much higher than those reported for graphene surface electrochemical

studies. The high currents are attributed to a combination of the nanopore edge structures produced by electron beam sculpting along with the convergent diffusion mechanisms due to nanosized electrodes, which have been reported to enhance ionic flux of reactive species. We also demonstrated the modulation of ionic current by the use of the embedded conductive

graphene terminal. Numerical simulations were performed to confirm the transistor like characteristics of the device. Extremely high electrochemical current densities have exciting applications for both chemical and biological sensing as well as energy storage. The scaling of these structures by producing arrays of nanopores could enable multiple applications.

## EXPERIMENTAL DETAILS

**Graphene Growth and Transfer.** Graphene is grown by chemical vapor deposition (CVD) on 1.4 mil copper foils purchased from Basic Copper.<sup>21,24,55</sup> Copper foil is placed in an Atomate CVD system and annealed at  $\sim 1000^\circ\text{C}$  under Ar/H<sub>2</sub> flow for 90 min at a base pressure of  $\sim 4.4$  Torr. Graphene is grown for 30 min at  $1000^\circ\text{C}$  under 850 sccm of CH<sub>4</sub> and 50 sccm of H<sub>2</sub> at a base pressure of about 2.5 Torr. The resulting graphene and Cu substrates are cooled to  $400^\circ\text{C}$  under 850 sccm of CH<sub>4</sub>, 50 sccm of H<sub>2</sub> at a rate of  $\sim 10^\circ\text{C}/\text{minute}$  followed by cooling to room temperature under 500 sccm of Ar while the base pressure is ramped to 760 Torr (Supporting Information, Figure S8a). Graphene is transferred to the receiving substrates by coating one side of the Cu foil with a bilayer of PMMA (495 K A2 and 950 K A4) (Supporting Information, Figure S8b-i). Each layer of PMMA is coated at 3000 rpm followed by a  $200^\circ\text{C}$  bake for 2 min. The backside graphene is removed by O<sub>2</sub> plasma etching prior to etching the Cu foil (Supporting Information, Figure S8b-ii) in etchant overnight (Transcene CE-100). The resultant PMMA/graphene film is transferred to a 10% HCl in deionized (DI) water solution to remove residual metal particles followed by a second DI rinse (Supporting Information, Figure S8b-iii). The film is then transferred onto the receiving substrate (Supporting Information, Figure S8b-iv) with predefined FIB holes ( $\sim 300$  nm in diameter) and PMMA is removed in a 1:1 methylene chloride/methanol solution for 30 min. The samples undergo a  $400^\circ\text{C}$  anneal under Ar (500 sccm) and H<sub>2</sub> (100 sccm) flow to remove residual PMMA.

**Raman Spectroscopy and AFM Characterization.** Raman mapping is performed using a scanning confocal Renishaw Raman microscope (inVia and WiRE 3.2 software). Data are collected using a 633 nm edge emitting laser (laser spot size  $\sim 1.3\ \mu\text{m}$  and  $\sim 0.1$  mW incident power), a  $50\times$  long working distance objective, a 1800 lines/mm grating, and 30 s acquisition time; 121 spectra are collected over a  $20 \times 20\ \mu\text{m}^2$  area at a  $2\ \mu\text{m}$  step size and analyzed by fitting mixed Gaussian and Lorentzian curves around the D, G, and 2D Raman peaks centered at *ca.* 1340, 1590, and  $2660\ \text{cm}^{-1}$ , respectively. A cubic spline interpolation is used to subtract the background before curve fitting. Atomic force microscope (AFM) data are collected using a Digital Instruments Dimension 3000 AFM in tapping mode. Calculated root-mean-square (RMS) roughness values are obtained using Nanoscope Analysis v.1.4 software from Bruker Corporation. Three dimensional images are rendered using Gwyddion AFM analysis software.

**Supporting Membranes.** Membranes consisting of stacked layers of Al<sub>2</sub>O<sub>3</sub> and SiN<sub>x</sub> are fabricated on  $300 \pm 2\ \mu\text{m}$  thick double-side polished (100) silicon wafers purchased from Silicon Quest International. Wafers are piranha cleaned (1:1 H<sub>2</sub>SO<sub>4</sub>/H<sub>2</sub>O<sub>2</sub>) for 15 min before depositing Al<sub>2</sub>O<sub>3</sub> via ALD (Cambridge Nanotech). Al<sub>2</sub>O<sub>3</sub> (50 nm) was deposited at a platen temperature of  $250^\circ\text{C}$  using tetramethyl-aluminum (TMA) and water vapor precursors. Subsequently, 200 nm of low-stress SiN<sub>x</sub> is deposited (STS Mesc PECVD system) using a mixed-frequency recipe (high frequency, 6 s at 13.56 MHz, platen power of 20 W; and low frequency, 2 s at 380 kHz, platen power of 60 W) with precursors SiH<sub>4</sub> and NH<sub>3</sub> at flow rates of 40 and 55 sccm, respectively, at a platen temperature of  $300^\circ\text{C}$ . Another 50 nm of Al<sub>2</sub>O<sub>3</sub> (ALD) is deposited with the same parameters as described before. Optical lithography is used to define  $80\ \mu\text{m}$  square windows on the back of the wafer with the aid of plasma resistant Megaposit SPR-220 photoresist and an ABM

Flood Exposure (model 60) tool. The wafer is then placed inside an STS Pegasus ICP DRIE and  $80\ \mu\text{m}$  square membranes are suspended using a Bosch etching process; 300 to 350 nm holes are then formed in these membranes using a focused ion beam (FIB) (FEI DB235) operated at a beam current of 30 pA.

**Nanopore Fabrication and Nanopore Fluidic Measurement.** The graphene–Al<sub>2</sub>O<sub>3</sub>–graphene–Al<sub>2</sub>O<sub>3</sub> stack is fabricated sequentially using the same graphene transfer and ALD process as described previously. The thickness of Al<sub>2</sub>O<sub>3</sub> for both dielectric layers is 24 nm. An Al seed layer (2 nm thick) is deposited on graphene using a CHA SEC-600 electron-beam evaporator prior to deposition of both dielectric layers. The second (top) graphene layer is contacted with Ti/Au contacts. Electrical contacts, Ti (2 nm thickness adhesion layer) and Au (300 nm thick), are deposited onto G2 by shadow masking and e-beam evaporation. The measured sheet resistance of graphene is  $6.7\ \text{k}\Omega/\square$ . Single nanopores of 5–20 nm diameter are drilled in the graphene-embedded membrane using a JEOL 2010F field-emission gun TEM operated at 200 kV in CBED mode with focused electron probe of diameter = 1.6 nm. O<sub>2</sub> plasma treatment at 50 W for 30 s on source side facilitates wetting. Subsequently Al wires are attached on Ti/Au contacts using silver paint and the chip is assembled in a custom-built chamber. Ethanol is filled in both reservoirs initially to promote wetting. Subsequently the ethanol is flushed out and the reservoirs are filled with a solution of 1 M KCl, 10 mM Tris, 1 mM EDTA at pH 7.6. All nanopore experiments are performed with Axopatch 200B and Digidata 1440A at room temperature ( $22 \pm 2^\circ\text{C}$ ).

**Electrostatic Simulations.** The mathematical model for ion transport involves a set of equations governing ionic transport and the electric potential.<sup>56</sup>

The total flux due to diffusion and electromigration of the *i*th species (ions) is given by the following expression

$$\Gamma_i = -D_i \nabla c_i - \frac{D_i}{RT} z_i F c_i \nabla \phi \quad (3)$$

where *F* is the Faraday's constant, *z<sub>i</sub>* is the valence, *D<sub>i</sub>* is the diffusion coefficient,  $\Gamma_i$  is the flux, *c<sub>i</sub>* is the concentration of the *i*th species, and  $\phi$  is the electrical potential. The Nernst–Planck (NP) equation describes the reaction rate (*r<sub>i</sub>*) of dissolved species.

$$\frac{\partial c_i}{\partial t} = -\nabla \cdot \Gamma_i + r_i \quad (4)$$

The electrical potential distribution is governed by the Poisson equation

$$\nabla \cdot (\epsilon_r \nabla \phi) = -\frac{F \sum_i z_i c_i}{\epsilon_0} \quad (5)$$

where  $\epsilon_0$  is the permittivity of vacuum and  $\epsilon_r$  is the relative permittivity. The electric potential at the wall surface is governed by

$$\frac{\partial \phi}{\partial n} = -\frac{\sigma_s}{\epsilon_0 \epsilon_r} \quad (6)$$

where  $\sigma_s$  is the surface charge density and *n* is the normal to the wall.

The Poisson–Nernst–Planck equations (PNP) equations can be simplified by integrating eq 3 and 5 across the channel,

which gives

$$\frac{\partial \bar{c}_i}{\partial t} = \frac{1}{A} \frac{\partial}{\partial x} \left( AD_i \frac{\partial \bar{c}_i}{\partial x} + A \frac{D_i}{RT} z_i F \bar{c}_i \frac{\partial \bar{\phi}}{\partial x} \right) + \bar{r}_i \quad (7)$$

and

$$\frac{\partial}{\partial x} \left( A \frac{\partial \bar{\phi}}{\partial x} \right) = - \frac{A(F \sum z_i \bar{c}_i + 4\sigma_s/d)}{\varepsilon_r \varepsilon_0} \quad (8)$$

where  $A$  is the cross-sectional area,  $x$  is the coordinate along the channel,  $d$  is the nanopore diameter,  $\bar{c}_i$ ,  $\bar{r}_i$ , and  $\bar{\phi}$  are the cross-sectional averaged concentration, reaction rate, and electric potential, respectively.

From eq 7 and 8, we obtain the cross-sectional averaged electric potential, and ionic concentration. The drain current is calculated by multiplying the current density along the  $x$  direction (assumed normal to pore wall) with the cross-sectional area at drain.

$$I_x = \sum_i z_i F \Gamma_{xi} \quad (9)$$

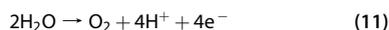
where  $\Gamma_{xi}$  is the flux rate of  $i$ th species in the  $x$  direction.

The gate current is calculated from the reaction rate of the species near the graphene gate. The oxidation rate of  $\text{Cl}^-$  is assumed as a function of the electrical potential bias and the local concentration.

$$\bar{r}_{\text{Cl}^-} = -r_{\text{Cl}^-} [\exp(aV_G - bV_D) - 1] \bar{c}_{\text{Cl}^-} \quad (10)$$

where  $r_{\text{Cl}^-}$ ,  $a$ , and  $b$  are fitting parameters.  $\bar{c}_{\text{Cl}^-}$  is the cross-sectional averaged concentration of  $\text{Cl}^-$  at pore surface (mM or mol/m<sup>3</sup>).

Near the graphene gate edge, water oxidation generated  $\text{H}^+$  ions:



The generation rate of  $\text{H}^+$  is assumed as

$$\bar{r}_{\text{H}^+} = -r_{\text{H}^+} [\exp(aV_G - bV_D) - 1] \quad (12)$$

The gate current  $I_G$  is calculated from

$$I_G = \frac{\pi}{4} d^2 l_r F (\bar{r}_{\text{H}^+} - \bar{r}_{\text{Cl}^-}) \quad (13)$$

where  $l_r$  is the length of reaction region. In the simulations,  $l_r$  is taken as 2 nm,  $r_{\text{Cl}^-} = 5 \times 10^7 \text{ s}^{-1}$ , and  $r_{\text{H}^+} = 2 \times 10^7 \text{ mol}/(\text{m}^3/\text{s})$ . The parameters  $a$  and  $b$  are correlated to the pore size and electrolyte concentrations. In these simulations,  $d > \kappa^{-1}$  where  $\kappa^{-1}$  is the Debye layer thickness, we choose  $a = 0.66 \text{ V}^{-1}$ , and  $b = 2.05 \text{ V}^{-1}$ . However when  $d < \kappa^{-1}$ , for  $d = 5 \text{ nm}$  and KCl concentration of 10 mM, we choose  $a = 0.625 \text{ V}^{-1}$  and  $b = 1.25 \text{ V}^{-1}$ . The gate current from water oxidation is much smaller than that from  $\text{Cl}^-$  oxidation. However water oxidation induces  $\text{H}^+$  ions in the nanopore, which affects the surface charge density of the  $\text{Al}_2\text{O}_3$  layer.

The surface charge density of the  $\text{Al}_2\text{O}_3$  layer is determined by the density difference of the sites attracting positive and negative charges.

$$\sigma_s = e(N_+ - N_-) \quad (14)$$

$$N = N_+ + N_- + N_0 \quad (15)$$

where  $N_+$ ,  $N_-$ , and  $N_0$  are the density of positively charged, negatively charged, and neutral sites, respectively.

The densities of the positively and negatively charged sites are related to the pH value and surface potential  $\psi_s$ .<sup>57</sup>

$$\frac{N_+ N_0^{\text{ISP}}}{N_+^{\text{ISP}} N_0} = \frac{\bar{c}_{\text{H}^+}}{c_{\text{H}^+}^{\text{ISP}}} \exp\left(-\frac{F\psi_s}{RT}\right) \quad (16)$$

$$\frac{N_- N_0^{\text{ISP}}}{N_-^{\text{ISP}} N_0} = \frac{c_{\text{H}^+}^{\text{ISP}}}{\bar{c}_{\text{H}^+}} \exp\left(\frac{F\psi_s}{RT}\right) \quad (17)$$

where ISP is the isoelectric point,  $c_{\text{H}^+}$  is the  $\text{H}^+$  concentration in the nanopore. The surface charge density is obtained by the

Grahame equation.<sup>34</sup>

$$\sigma_s = \varepsilon_r \varepsilon_0 \frac{RT}{zF} \kappa \sinh\left(\frac{F\psi_s}{2RT}\right) \quad (18)$$

Given  $N$ ,  $c_{\text{H}^+}^{\text{ISP}}$ , and  $N_0^{\text{ISP}}$ , the surface charge density can be obtained by solving eqs 14–18. In the simulations,  $c_{\text{H}^+}^{\text{ISP}}$  is chosen as  $10^{-8} \text{ mM}$ ,  $N = 6/\text{nm}^2$ , and  $N_0^{\text{ISP}} = 2/\text{nm}^2$ . We calculate  $\bar{c}_{\text{H}^+} = 0.01 \text{ mM}$  in a 5 nm pore (for  $V_{\text{ds}} = -500 \text{ mV}$  and  $V_{\text{gs}} = 500 \text{ mV}$ ). We also calculate  $\bar{c}_{\text{Cl}^-} = 5000 \text{ mM}$  and  $\bar{c}_{\text{Cl}^-} = 8500 \text{ mM}$  for 10 mM and 1000 mM KCl solutions, respectively.

**Conflict of Interest:** The authors declare no competing financial interest.

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**Supporting Information Available:** Leakage studies, additional graphene electrochemical studies, photographs of experimental setup, characterization of graphene transfer process and dielectric deposition. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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